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Interpenetrating Composites with Enhanced Stiffness, Desired Poisson's ratio and Superior Conductivity

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Summary

Fibre matrix debonding, fibre pullout, delamination and mechanical anisotropy are the main common disadvantages of most fibre-reinforced composites. Interpenetrating phase composites (IPCs), however, do not have these problems because both their matrix material and their reinforcement fibre materials are self-connected networks, and interpenetrate each other. Moreover, IPCs could be designed to have an almost isotropic Young's modulus much larger than the Voigt limit, and a Poisson's at a desired value (i.e. positive, or negative or zero). In addition, they could have an isotropic thermal or electrical conductivity very close to the theoretical upper limit (i.e. the Hashin—Shtrikman's upper limit). This paper will introduce the relevant theoretical, simulation and experimental results on the elastic properties and thermal/electrical conductivities of some IPCs and compare their properties with those of other types of composites.

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Keywords: Interpenetrating phase composites, Elastic properties, Poisson's ratio, Conductivities.

1 Introduction

2 Fibre composites are widely used in engineering 3 applications, in which the fibres are usually much 4 stronger and stiffer than the matrix material and used to reinforce the matrix material. To make the best 5 use of the reinforcement material, it is very 6 7 important that the reinforcement material is self-8 connected to form a network structure. For example, 9 in modern buildings, bridges or water containing 10 dams, the reinforcement steel bars are welded 11 together to form a self-connected network, and the 12 concrete material (i.e. the matrix material) is then 13 casted into the porous space of the steel network. If 14 the reinforcement steel bars are not self-connected, 15 even if the same amount of the reinforcement steel material is used in buildings, bridges or dams, these 16 17 structures could easily fall apart. Thus, in 18 conventional fibre composites, the reinforcement fibres are in general not best used. Their common 19 disadvantages include fibre matrix debonding, fibre 20 21 pullout, delamination and mechanical anisotropy, 22 [1]. In contrast, interpenetrating phase etc 23 composites (IPCs) can avoid all these disadvantages. 24 By theoretical analysis and computational 25 simulations, Zhu et al. [2,3] and Zhang et al. [4,5] 26 have found that IPCs could have an almost isotropic 27 Young's modulus much larger than the Voigt limit, 28 a Poisson's ratio at a desired value (i.e. positive,

negative or zero), and a thermal or electrical
conductivity close to the Hashin-Shtrikman's upper
limit[6]. The aim of this paper is to highlight the
advantages of IPCs over the fibre or particle
composites.

34 Material Models

In our theoretical and simulation research works, the 35 36 IPCs are reinforced by a self-connected periodic 37 regular fibre network or a lattice structure. Thus, 38 periodic representative volume elements (RVEs) 39 and periodic boundary conditions can be used to 40 obtain the elastic properties and thermal/electrical 41 conductivities. Based on the geometric feature of the 42 reinforcement network structure, the IPCs are 43 classified into two main types: normal and auxetic. 44 In the normal IPCs as shown in Fig. 1, the 45 reinforcement fibre network is a normal network or 46 lattice which has a positive Poisson's ratio. In the 47 auxetic IPCs as shown in Fig. 2, the reinforcement 48 fibre network is an auxetic network or lattice which 49 has a negative Poisson's ratio. In theoretical 50 analysis, the reinforcement fibre network and the 51 matrix are divided into a number of blocks. In finite 52 element simulations, both the reinforcement fibre 53 network and the matrix are partitioned into a large 54 number of tetrahedral elements.

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11 **Results**

13 A. Enhanced Young's modulus

14 The Voigt limit has long been regarded as the upper 15 limit for the Young's moduli of isotropic composites 16 [1]. For composites composed of a reinforcement 17 material with a Young's modulus of E_f and a 18 matrix material with a Young's modulus of E_m , the

19 Voigt limit is given as

$$20 \qquad E_{Voigt} = E_f V_f + E_m V_m \tag{1}$$

21 where V_f and V_m are the volume fractions of the 22 fibre and matrix materials in the composites, 23 respectively, and $V_f + V_m = 1$.

To make the theoretical and the finite element 24 25 simulation results of the IPC more useful, the Young's modulus of the matrix material is assumed 26 to be 1, and the Young's modulus of the 27 reinforcement fibre material is the value of 28 29 E_f/E_m , i.e. the ratio of the actual Young's 30 modulus of the fibre material to that of the matrix material. Further, the obtained Young's of the IPC 31 32 is normalized by the Voigt limit given by Eq. (1). 33 For different types of IPCs reinforced by a normal 34 fibre network shown in Fig. 1, the effects of the

35 different combinations of the constituent material

36 properties on the normalized Young's moduli of the

37 IPCs are presented in Fig. 3.

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Fig. 3. Effects of the different combinations of the constituent material properties on the normalized Young's modulus of different types of normal IPCs [5], where v_f and v_m are the Poisson's ratios of the fibre and matrix materials, respectively.

46 Both the constituent materials, i.e. the fibre and 47 matrix, are assumed to be isotropic. Thus, the 48 possible range of their Poisson's ratio is (-1.0, 0.5). 49 The theoretical [2] and finite element simulation [5] 50 results indicate that the elastic properties of the IPCs 51 are nearly isotropic. The results in Fig. 3(c) and 3(d) 52 show clearly that the Young's moduli of the IPCs 53 could be much larger than the Voigt limit, and 54 therefore much larger than those of the conventional 55 particle or fibre composites. In general, the larger the 56 difference between the Poisson's ratios of the matrix 57 and the reinforcement fibre materials, or the smaller 58 the difference between the Young's moduli of the 59 two constituent materials, the larger will be 60 normalized Young's modulus of the IPCs. It is noted that for different types of IPCs [4] reinforced by an 61 auxetic fibre network shown in Fig. 2, their Young's 62 63 moduli are in general smaller that those of the IPCs reinforced by a normal fibre network, but still much 64 larger than those of the conventional particle or fibre 65 reinforced composites. It is also worth noting that 66 for the same given amounts of the matrix and 67 reinforcement materials, if the geometric structure 68 69 of the reinforcement material is a perfect regular 70 closed cell foam with a uniform wall thickness and the matrix material fills the identical cubic cells, the 71

resultant composite [7] has the largest nearly 1 2 isotropic Young's modulus. This is because among all the possible geometrical structures of the self-3 4 connected porous reinforcement material, the 5 regular closed-cell foam structure with identical 6 cubic cells and uniform wall thickness has the 7 largest nearly isotropic stiffness. However, this type of composites is not IPC because its matrix 8 9 material/phase doesn't form a self-connected 10 network.

11 B. Desired value of Poisson's ratio

12 For different types of IPCs with different fibre 13 volume fractions and reinforced by a normal fibre 14 network shown in Fig. 1, the effects of the different combinations of the constituent material properties 15 on the Poisson's ratio of the IPCs are demonstrated 16 17 in Fig. 4. As can be seen, if both the matrix and the 18 fibre materials have a positive Poisson's ratio, the 19 Poisson's ratio of the IPCs would always be positive 20 [5]. If the matrix material has a large magnitude of negative Poisson's ratio, the Poisson's ratio of the 21 IPCs could have a large magnitude negative 22 23 Poisson's ratio. 24



Fig. 4. Effects of the different combinations of the constituent material properties on the

28 Poisson's ratio of different types of normal IPCs

29 [5] reinforced by a normal fibre network.

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30 For different types of IPCs with different fibre 31 volume fractions and reinforced by an auxetic fibre network shown in Fig. 2, the effects of the different 32 combinations of the constituent material properties 33 34 on the Poisson's ratio of the IPCs [4] are 35 demonstrated in Fig. 5. In contrast to the results of the IPCs reinforced by a normal fibre network, even 36 if the Poisson's ratios of both the matrix and the fibre 37 materials are positive, the IPCs reinforced by an 38 39 auxetic fibre network can have a large magnitude 40 negative Poisson's ratio. 41 Based on the results demonstrated in Figs 4 and 5, it

42 is concluded that the Poisson's ratio of PCs could be 43 designed to achieve a desired value (i.e. positive, or 44 negative or zero) by carefully choosing the45 combination of the properties of the constituent46 materials.

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Fig. 5. Effects of fibre volume fraction on the Poisson's ratio of the composites [4] when $\alpha = 20^{\circ}$. (a) $v_m = 0.1$, $v_f = 0.25$, E_f / E_m = 1000; (b) $v_m = 0$, $v_f = 0.25$, $E_f / E_m = 1000$.

54 C. Superior conductivity

55 In two phase composites, the constituent fibre and 56 matrix are assumed to be homogenous and isotropic 57 materials A and B, with conductivities μ_A and μ_B

and volume fractions V_A and V_B , respectively. 59

60 For composites with anisotropic conductivity, the largest and the smallest possible effective 61 conductivities can be easily achieved if the two 62 constituent materials A and B are uniformly 63 64 arranged in parallel, for example, sandwich/laminate 65 composites with layers of uniform thickness. For 66 such anisotropic composites, their upper limit of 67 conductivity is given by $\mu_U = \mu_A V_A + \mu_B V_B$ and their

- 68 lower limit is given as $\mu_L = \frac{\mu_A \mu_B}{\mu_A V_B + \mu_B V_A}$, where
- $\begin{array}{ll} 69 & V_{A} + V_{B} = 1 \, . \\ 70 & \end{array}$

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For composites with isotropic conductivity, the
magnitude of the conductivity is limited by the
Hashin and Shtrikman's upper and lower bounds [6].
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$$\mu_{HS_{-}U} = \mu_{A} + \frac{V_{B}}{\frac{1}{\mu_{B} - \mu_{A}} + \frac{V_{A}}{3\mu_{A}}}$$
(2)
$$\mu_{HS_{-}U} = \mu_{A} + \frac{V_{A}}{2\mu_{A}}$$
(3)

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$$\mu_{HS_{-L}} = \mu_B + \frac{V_A}{\frac{1}{\mu_A - \mu_B} + \frac{V_B}{3\mu_B}}$$
 (3)

77 where it is assumed that $\mu_A \ge \mu_B$.

78 For IPCs reinforced by the normal type-I fibre 79 network with $V_A = 0.104$, the relationship between 80 the effective conductivity and the ratio μ_A / μ_B has

81 been obtained by theoretical analysis and finite 82 element simulation using ABAQUS [3]. Fig. 6 83 shows the effects of the ratio μ_A / μ_B on the 84 effective conductivity of such IPCs, where different

- 1 bounds/limits are plotted for comparison [3] and the
- 2 results are normalized by μ_B . As can been seen, the

3 theoretical results are very close to those of the finite

4 element simulation results, and the conductivities of

5 the IPCs are closer to the Hashin- Shtrikman's upper

6 bound than to their lower bound (see Fig. 6), and much7 larger than the experimentally measured results of the

8 conventional particle or short- fibre composites, as

9 shown in Fig. 7.



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11 **Fig. 6.** Effects of μ_A / μ_B on the conductivity

12 of IPCs reinforced by a normal type-I fibre 13 network [3], where the fibre volume fraction 14 $V_A = 0.104$.



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Fig. 7. Comparison between the conductivities of
IPCs and the experimentally measured results of
the conventional particle and short-fibre
composites, where the results are normalized by
the Hashin-Shtrikman's upper limit.

21 It is noted that for the same given amounts of the 22 matrix and reinforcement materials, if the geometric 23 structure of the reinforcement material is a perfect regular cubic closed cell foam with a uniform wall 24 25 thickness and the matrix material fills the identical 26 cubic cells, the resultant composite will have the 27 largest isotropic conductivity [7], which is exactly 28 the same as the Hashin-Shtrikman's upper limit. 29 However, such composite is not an IPC.

30 Discussion and Conclusions

31 It is relatively easy to manufacture IPCs. The 32 regular reinforcement fibre network structure can be 33 produced first, the matrix material (e.g. concrete, 34 resin, or polymer) can then be casted into the selfconnected porous network of the reinforcement 35 structure. The theoretical and finite element 36 37 simulation results have demonstrated that for the 38 same amounts of the constituent matrix and 39 reinforcement materials used, IPCs can have a much 40 larger nearly isotropic Young's modulus than those 41 of the conventional particle or short fibre composites. Further, the nearly isotropic Young's modulus of 42 43 IPCs could be designed to be much larger than the 44 Voigt limit that was generally regarded as the 45 unexceedable upper limit for all composites. In addition, IPCs can be designed as functional 46 47 material with a Poisson's ratio at a desired value, e.g. 48 positive, or negative, or zero. Moreover, IPCs have 49 a conductivity significantly larger than those of the 50 conventional particle and short-fibre composites. 51 Therefore, IPCs have very important engineering 52 applications in many different areas.

53 Conflicts of Interest

54 The authors declare no conflict of interest.

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